

Structural transformation in AI-51 at.% Zn eutectoid alloy observed in internal friction

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• Zn-Al based alloys have been commercially accepted for many years, since the first casting alloy was developed at the New Jersey Zinc Company in 1922

General Rule of Phase Decomposition in Zn-Al Based Alloy

Since the 1970's, systematic investigation of phase relationships in Zn-Al (ZA) based alloys Studies began with the establishment of phase diagrams of alloy systems (Zn-Al binary, Zn-Al-Cu, Zn-Al-Si ternary and Zn-Al-Cu-Si quaternary)

Followed by studies of the phase transformations which occurred in solution-treated alloys

Phase diagram of Presnyakov et al. modified by Goldak and Parr has been adopted for representation of Zn-Al binary phase diagram [Zhu²⁰⁰⁴]



[Zhu²⁰⁰⁴] Y.H. Zhu, 'General Rule of Phase Decomposition in Zn-Al Based Alloys (II) -On Effects of External Stresses on Phase Transformation-': Materials Transactions, 45(11) (2004) 3083-3097

[Presnyyakovu¹⁹⁶¹] A.A. Presnyyakov, Y.A. Goltban, V.C. Cherptyyakova: Russ. J. Phys. Chem. 35 (1961) 623-633

WHAT ABOUT AI-Zn PHASES DIAGRAM

• Zn-Al phases diagram 'light' to 'heavy' information



[Dartyge¹⁹⁶⁹] E. Dartyge, J.M. Dartyge, M. Lambert, A. Guinier, 'Étude de la résistivité d'une solution solide Al-Zn au cours de sa précipitation': Journal de Physique, 30(1) (1969) 82-86

[Željko²⁰⁰⁹] S. Željko, P. Stanko, S. Goran, 'Microstructure of Al-Zn and Zn-Al Alloys': Croatica Chemical Acta 82(2) (2009) 405-420

WHAT ABOUT AI-Zn STUDIES & microSTRUCTURES

Among these materials, the eutectoid alloy AI 51%.at Zn

At the microstructural level, observations by different authors [Zhu¹⁹⁹⁷] [Zhu²⁰⁰⁴] [Hirata¹⁹⁹⁸] [Zhang²⁰⁰⁶] on the eutectoid alloy in particular showing that the configuration of the mixture of the two phases α and β is very complex, thermo and mechanical dependent

Lamellar structures, deformed lamellar structures, globular structures, interconnected globular structures and clusters.

In addition, different aspects (chemical composition, phase evolution, etc.) & try to identify the transition from one structure to another [Zhu¹⁹⁹⁷] [Hirata¹⁹⁹⁸] [Zhu²⁰⁰⁴] However, no statistical study established to indicate the proportion of each 'state'



TEM microstructures in Zn–Al eutectoid alloy after solution treatment at 360C-6hrs, followed by furnace cooling. $\beta^{-(dark plate)}$ and $\alpha^{-(bright plate)}$ phases ; (a) Straight lath, and (b) curved lath [Zhang²⁰⁰⁶] *M.-X. Zhang, P.M. Kelly, 'Understanding the crystallography of the eutectoid microstructure in a Zn-Al alloy using the edge-to-edge matching model': Scripta Materialia 55* (2006) 577-580

[Hirata¹⁹⁹⁸] V.M.L. Hirata, M. Saucedo Munoz J.C. Rodriguez Hernandez Y.H. Zhu, 'Milling characteristics of extruded eutectoid Zn–Al alloy': Materials Science and Engineering A247 (1998) 8-14

[Zhu¹⁹⁹⁷] Y.H. Zhu, V.M.L. Hirata, M.S. Munoz, 'Milling induced microstructural change in furnace cooled eutectoid Zn single bond Al alloy': Journal of Materials Processing Technology 63 (1997) 624-627

1. Introduction

State of the art, thermal behavior & structural transitions in Al-Zn alloys
 Isothermal Mechanical Spectroscopy *versus* Structure Analysis

2. Forced pendulum versus SEM analysis

At constant temperature, At low frequency Activation parameters Structure Analysis: **A**(lamellar) to **E**(globular)

3. Physical Analysis in Al-51 at.% Zn alloy

Stabilised structures & Spectroscopy spectra

4. Discussion/Conclusion

[Rivière²⁰¹²] A. Rivière, V. Pelosin, M. Gerland, 'Eutectoid Al-51at%Zn Alloy Studied by Isothermal Mechanical Spectroscopy': Solid State Phenomena 184 (2012) 167-172
[Zhu¹⁹⁹⁰] X.F. Zhu, 'Stable damping associated with linear viscous motion of the interface in a multiphase Al-Zn alloy': Journal of Applied Physics 67 (1990) 7287 (1990)

THE FORCED PENDULUM: Temperature & Low Frequency

Stress-Strain Dephasage Measurement





The moving part is suspended by four very fine crossed ribbons to prevent any possibility of flexion

⇒ Only isothermal experiments

Characteristics for internal friction experiments:

- Frequency range: 10⁻⁵ Hz 50 Hz
- Temperature range: 300 K 1400 K
- Maximal strain: 2×10⁻⁷ 5×10⁻⁶ 2×10⁻⁵
- Damping accuracy: 10⁻⁴

[Rivière²⁰⁰¹] A. Rivière, in: R. Schaller, G. Fantozzi, G. Gremaud (Eds.): 'Mechanical Spectroscopy Q⁻¹ 2001 with Applications to Materials Science', Trans Tech Publication Ltd., Switzerland, 2001, p. 635,

Materials & Procedure: SEM analysis

Binary AI-51 at.% Zn, eutectoid alloy (Goodfellow Company)

- ✓ Let's fix and observe: Figs. 1 & 2 with the general appearance on quenched sample (first treatment 653K) from 385 and 493K, 2hrs30min holding time for each T_{annealing}
- Presence of "greyish" zones more preponderant in 385K sample compare to 493K;
 Fig.1 lamellar structure predominates, *Fig.2* globular structure, which is dominant



Fig.1 AI-51% at. Zn, quenched from 385K

SEM-FEG (JEOL 7000 Scanning Electron Microscopy-Field Emission Gun operating at 7 kV)



Fig.2 Al-51% at. Zn, quenched from 493K

AI-51 at.% Zn: SEM vs. FDIF methodologies

- Evolution of the two structures, SEM analyzed into small areas segmentation:
- A, lamellar structure (greyish)
- B, destruction of the lamellar structure and progressive evolution towards the globular one
- C, formation of the globular structure
- D & E, respectively, beginning and end of the globules process coalescence

SEM size (1200 pixels in width) (900 pixels in height) divided into 885 square zones (3.95% of the total) of 35 pixels
Procedure counting the number of each species (A to E) for SEM images for each T: average of all photos from the different observation areas of the same sample



Fig.3 AI-51% at. Zn, successive elementary surfaces describing (A to E)

 SEM (385, 433 and 493K - 2hrs30min) chosen in T range of the internal friction peaks evolution FDIF
 Frequency dependent

internal friction

- Annealing (24hrs, vacuum 10⁻³Pa)

- Temperatures steps Atm to 540K
- Deformation

max. 5.10⁻⁶

- Frequency 10⁻⁵- 50Hz

SEM STRUCTURE ANALYSIS: AI-51 at.% Zn, 385K

- Statistically detection of 854 surfaces presented in the form of a histogram (*Fig.4*)
- B, corresponding to the destruction of the lamellar structure and its progressive transformation into a globular structure, is preponderant
- > This change from A to B necessarily induces a change in the shape of the interface between α and β phases



Fig.4 AI-51% at. Zn, quenched from 385K



Fig.4 AI-51% at. Zn, 385K: SEM analysis

SEM STRUCTURE ANALYSIS: AI-51 at.% Zn, 433K

- Statistically detection of 794 surfaces, histogram (*Fig.5*)
- All the area A to E exist
- However, predominant areas B and D with practically an equal importance: correspond respectively to the destruction of the lamellar and globular structure



Fig.5 AI-51% at. Zn, quenched from 433K



Histogram AI-51% at. Zn, 433K

Fig.5 AI-51% at. Zn, 433K: SEM analysis

SEM STRUCTURE ANALYSIS: AI-51 at.% Zn, 493K

- Statistically detection of 832 surfaces presented in the form of a histogram (*Fig.6*)
- Significant increase in D and E surfaces categories, confirming the total transition from the lamellar structure to the coalescence of the globular structure
- Lamellar structure completely disappears, giving way to the coalescence of the globular structure



Fig.6 AI-51% at. Zn, quenched from 493K



Histogram Al-51% at. Zn, 493K

Fig.6 AI-51% at. Zn, 493K: SEM analysis

P1 and P2 closely related to the microstructural transformation?

Review: Frequency dependent Internal Friction Spectra at temperature range 385-467K, during first heating

- Below the eutectoid transition temperature, the structure is a mixture of the two α-β phases
- IF (Fig.7) revelation of two relaxation peaks P1 and P2, where with increasing T>350K:
- P1 shift to higher frequencies with a decrease in its amplitude
- P2 appearance at higher frequencies and progressive increase

Behavior closely related to evolution of the microstructure obtained under the eutectoid transition ?



*Fig.***7** Low frequency Internal Friction Spectra, first heating at various measurement T

AI-51 at.% Zn PHYSICAL ANALYSIS: Activation energy

P1 and P2 thermally activated Activation parameters during T decrease, 467, 527 and 544K $H_z = 0.75eV$, $\tau_0 = 10^{-7}$ > Close to Zhu, Al-60 at % Zn

exp. in (290-390K) at (0.1 and 7Hz)

- [Zhu¹⁹⁹⁰] observed damping because thermally induced atom diffusion across the interface between α and β phase grains during decreasing measurement T
- [Rivière²⁰¹²] hypothesis to explain P1 and P2 co-existence, believes that the progressive 'P1 disappearance' and simultaneous 'P2 appearance & increase' could correspond to a change in the shape of the lamellar interface between α and β ? Together with atoms diffusion at the interface.



Fig.8 Napierian logarithm of frequency of peak maxima plotted *versus* inverse measurement T, first heating and after various annealing

AI-51 at.% Zn: FI versus Structures

Evolution of the microstructure obtained under the eutectoid transition

SEM observations, to highlight the close link between the existence of the two peaks and the evolution of the microstructure
Fig.7 IF, first heating, variable of the microstructure

P1 FI-SEM-385K

 B preponderance, progressive lamellar structure destruction

Going in same direction as P1 evolution, whose relaxation strength gradually decreases with T increase





Al-51 at.% Zn: Fl versus Structures

P1 FI-385K / Confirmation during FI-SEM-433K

- SEM B and D in equal importance: respectively gradual destruction of lamellar to globular structure and beginning of coalescence of the globular structure
- *Agreement with P1 and P2 relaxation strengths, amplitude of the peaks practically equal







P2 FI-467K & SEM-493K

- D and E, corresponding to destruction of the globular structure predominate
- Development of P2 to the detriment of P1
- Behaviour globally reflects continuous change in shape of interface between α and β phases with their shape evolution ; Zhu¹⁹⁹⁰ explains by the thermally induced atomic diffusion across the interface between the α and β phase grains





> Arrhenius (*Fig.8*)

Relaxation parameters seems to prove that we are in the presence of the same mechanism, thermal activation energy H is equal to 0.75eV (70kJ/mol) for both P1 and P2 with different microstructures

Thermally induced atoms diffusion across the α - β interface: indeed, fewer atoms involved in the elementary mechanism





Fig.8 Napierian logarithm of frequency of peak maxima plotted *versus* inverse measurement T, first heating and after various annealing

CONCLUSION: ANALYSIS on AI-51 %.at Zn & Eutectoid

- I. AI-51 at.% Zn alloy shows a total transformation of the eutectoid mixture at this %
- Ideal for studying the evolution of the alpha and beta phases as a function of temperature
- **II.** Correlation between microstructure evolution (SEM) and isothermal mechanical spectroscopy (IMS)
- IMS carried out over wide frequency range (10⁻⁵- 50 Hz) vs. T_{amb} to 540 K
- Upon heating, two peaks P1 and P2 below the eutectoid transition T_{eut} (550 K)
- Both thermally-activated ; P1 decreases and disappears with increasing T, while P2 appears and increases continuously until the eutectoid transition temperature
- (P1 essentially, 383K) (Coexistence P1/P2 same amplitude, 433K) (P2 essentially, 467K)
- **SEM** after quenches at (385, 433 and 493 K holding time of 2hrs30min)
- P1 peak associated to lamellar structure whose destruction leads to the collapse of P1
- P2 peak associated to coalescence of globular structure and as this coalescence the amplitude of this peak becomes more important
- > Associated with thermally induced diffusion of atoms across the α - β interface ; microstructure from a lamellar form of the β -phase that gradually transforms according to the areas A and B until the globular form

III. SEM highlighting transformation mechanisms by specifying structure change and collapse

19th International Conference on Internal Friction and Mechanical Spectroscopy 27.06 - 01.07.2022







19th International Conference on Internal Friction and Mechanical Spectroscopy

The Conference, originally planned as hybrid online and face-to-face event in Moscow (http://icifms19.ru), will take place as a virtual event based in Rome 27.06-01.07.2022

THANK YOU FOR YOUR KIND ATTENTION

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